

1 **Capturing sclera anisotropy using direct collagen fiber models:**
2 **Linking microstructure to macroscopic mechanical properties.**

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44 **Abstract**

45 Collagen fibers are essential to the mechanical behavior of soft tissues, including sclera.
46 Conventional models often represent these fibers statistically, potentially missing crucial aspects
47 of their role in tissue behavior. In this study, we expand on a direct fiber modeling approach that
48 we recently presented based on explicitly representing the sclera long, interwoven fiber bundles.
49 Specifically, our goal was to capture specimen-specific 3D fiber architecture and anisotropic
50 mechanics of four ovine sclera samples (superior from Eye-1, temporal and superior from Eye-2,
51 and temporal from Eye-3), each tested under five conditions: equibiaxial (1:1) and four non-
52 equibiaxial (1:0.75, 0.75:1, 1:0.5, and 0.5:1). Fiber architecture was extracted using polarized light
53 microscopy and reconstructed model fiber orientations agreed well with the histological
54 information (adjusted $R^2 > 0.89$). Material parameters were determined via inverse fitting to the
55 equibiaxial tests. Remarkably, the parameters obtained from equibiaxial fitting also accurately
56 predicted the mechanical response of the same sample under all four non-equibiaxial conditions.
57 This indicates that the models inherently captured tissue anisotropy through its fiber structure,
58 unlike conventional continuum models which require simultaneous multi-condition fitting. Our
59 findings support direct fiber modeling as a promising tool approach for linking tissue fibrous
60 structure and macroscopic mechanical behavior.

61

62 **1. Introduction**

63 Biomechanical and mechanobiological analyses are critical tools for understanding the
64 function and dysfunction of ocular tissues (Amini and Barocas 2009; Bellezza et al. 2000; Ethier
65 et al. 2004; Girard et al. 2015; Nguyen and Ethier 2015; Roberts and Liu 2017). In particular, the
66 biomechanics of the corneoscleral shell play a significant role in the pathophysiology of visually
67 impairing disorders such as glaucoma and myopia (Bianco et al. 2024; Feola et al. 2016; Grytz et
68 al. 2020; Midgett et al. 2020; Newson and El-Sheikh 2006). As with many other soft tissues in the
69 body, collagen fibers are the primary load-bearing component of the sclera, providing structural
70 integrity and mechanical strength to the tissue under various physiological and pathological
71 conditions (Boote et al. 2020; Girard et al. 2009b; Grytz et al. 2014; Jan et al. 2017a; Pijanka et
72 al. 2012). Hence, it is important to identify the biomechanical role of scleral collagen fibers, and
73 how their behavior relates to physiology and pathology (Coudrillier et al. 2012; Ethier et al. 2004;
74 Pijanka et al. 2012; Rada et al. 2006). In recent years, computational modeling has emerged as
75 a key approach to study the relationship between scleral fiber structure and its biomechanical
76 response, mainly because of the difficulty in accessing the eye, especially the posterior pole,
77 (Girard et al. 2009a; Grytz et al. 2014; Kollech et al. 2019; Pijanka et al. 2015; Voorhees et al.
78 2018; Whitford et al. 2016; Zhou et al. 2019).

79 In the field of collagenous tissue modeling, two predominant approaches are often employed:
80 phenomenological models (Bellini et al. 2011; Humphrey 2003; Khoiy et al. 2018b; Taber 2009)
81 and structurally faithful models (De Jesus et al. 2016; Lanir 1983; Mahutga et al. 2023; Taber
82 2009; Zhang et al. 2016). Phenomenological models, while capable of predicting large-scale
83 tissue-level deformations, fall short in providing insights into the mechanical environments at the
84 meso-scale. This is not a surprise or necessarily a failure given that, by construction, the models
85 are intended to work at larger scales. However, this limitation hampers their utility in studying the
86 mechanobiology of the extracellular matrix, particularly collagen.

87 Structurally faithful models offer the possibility of a more nuanced description to soft tissue
88 mechanics at smaller scales. For instance, Ragupathy and Barocas developed a structural model
89 that represented soft tissue mechanics as the sum of fiber contributions across various
90 orientations (Ragupathy and Barocas 2009). They formulated a closed-form analytic solution for
91 the integral describing the overall resultant stress of fibrous tissues, using a von Mises fiber
92 orientation distribution and an exponential fiber stress-strain relationship. Similarly, Fata et al.
93 employed a structural-based model to study postnatal growth in ovine pulmonary arteries (Fata

94 et al. 2014). Amini and colleagues demonstrated how collagen fibers deform in the clinically
95 relevant deformation of the glenohumeral capsule, using a purely kinematic approach without
96 solving stress balance equations (Amini et al. 2014). They utilized an in-plane convective
97 curvilinear coordinate system to capture the fiber deformation. While these studies provided
98 insights into changes in overall collagen fiber distributions, the fibers were not treated as discrete
99 load-bearing entities. Such an approach limits the structural insights solely to changes in overall
100 fiber distribution without offering specific details about individual fiber bundles.

101 Another important consideration regarding the abovementioned studies is that they assumed
102 affine deformations —meaning that the local deformation of fibers is assumed to directly follow
103 the global tissue deformation—which may not always accurately represent collagenous tissue
104 behavior. Conversely, non-affine deformations occur when fibers deform differently from the
105 overall tissue, due to local microstructural variations, interactions, or constraints (Chandran and
106 Barocas 2006; Lee et al. 2022). To address non-affine deformations, multi-scale approaches have
107 been employed to connect tissue-level stress with fiber-level stress and deformation through
108 volume averaging methods (Aghvami et al. 2016; Hadi et al. 2012; Mahutga et al. 2023;
109 Pakzadmanesh et al. 2024; Stylianopoulos and Barocas 2007). These models, however, often
110 rely on representative volume elements (RVE) and tend to overlook key characteristics of fibrous
111 constructions, such as fiber interweaving. This omission can limit the accuracy of these models
112 in capturing the full complexity of fibrous tissue behavior (Wang et al. 2020). Compared to RVE-
113 based approaches, the direct fiber modeling framework can naturally represent long-range strain
114 transmission through interwoven fibers, as recently demonstrated in (He et al. 2024).

115 In the case of sclera and optic nerve, accurately incorporating fiber characteristics at the
116 microscale is likely crucial for correct prediction of the biomechanical effects at the cellular and
117 axonal scales (Lee et al. 2022).

118 With the intent to improve the ability to model fibrous sclera, we recently introduced a direct
119 fiber modeling technique (Ji et al. 2023a). In this approach, fiber families, or bundles, were
120 reconstructed and simulated using detailed, specimen-specific fiber architectures derived from
121 histological analysis through polarized light microscopy (PLM) (Jan et al. 2015). In doing so, the
122 models account for fiber interweaving, fiber-fiber interactions, and long fibers. We demonstrated
123 the capabilities of the *direct fiber* modeling approach in two studies with different scales. In the
124 first study, we reconstructed a model of a small sample of temporal sclera, and then used an
125 inverse fitting approach to match experimental equi-biaxial stress-strain data from the literature,
126 simultaneously capturing the behavior in both radial and circumferential directions (Ji et al. 2023a).

127 In the second study, we applied the direct fiber modeling approach to reconstruct a model of the
128 optic nerve head (ONH) and the adjacent tissues, a substantially larger region than the sclera
129 sample (Islam et al. 2024). Due to the size and complexity of the region being modeled, we
130 adjusted our approach to utilizing histological data for reconstructing the model. Specifically, we
131 prioritized the modeling of larger fiber bundles, 200–400 μm thick, rather than smaller fibers.
132 Despite employing linear mechanical properties for the tissue, the model effectively captured the
133 overall behavior of the optic nerve head under inflation, including the emergence of nonlinear
134 characteristics (Islam et al. 2024). Subsequently, we demonstrated that the direct fiber model
135 outperforms the conventional continuum model in capturing both the large-scale tissue strain
136 pattern and the small-scale fiber strain in the sclera region of the ONH (He et al. 2024). While
137 both studies demonstrated the potential of the direct fiber modeling approach, they had several
138 important limitations. Firstly, the histological data and mechanical response data were obtained
139 from different specimens. Specifically, in the first study, the histological data were collected from
140 an ovine eye, whereas the mechanical response data came from a porcine eye. In the second
141 study, both sets of data were derived from the porcine ONH but from different porcine specimens.
142 Furthermore, only a single posterior sclera sample was analyzed, so it was not possible to confirm
143 that the method can consistently capture specimen-specific information. Finally, previous tests of
144 the model were limited to equi-biaxial loading conditions, which do not fully account for the more
145 complex, anisotropic forces that the sclera may encounter in physiological conditions. (Boote et
146 al. 2020).

147 Our goal in this study was to build on the promise of the studies mentioned above and continue
148 the development and evaluation of the direct fiber modeling of sclera. Specifically, we set a much
149 more challenging objective by requiring that the models capture specimen-specific 3D fiber
150 architecture and experimental anisotropic mechanics of scleral samples obtained from distinct
151 anatomical locations (superior quadrant of Eye-1, temporal and superior quadrants of Eye-2, and
152 temporal quadrant of Eye-3). In addition, we expanded the mechanical loading protocols, moving
153 from only equi-biaxial tests to a set of five tests including equi-biaxial and four non-equi-biaxial
154 tests, thereby covering a broader range of sample behavior and strain energy field.

155 We show that the direct fiber modeling approach effectively captured the non-linear and
156 anisotropic responses unique to each sample. Unlike phenomenological models, which often rely
157 on simultaneously fitting all biaxial testing protocols to the analytical model (Bellini et al. 2011;
158 Geest et al. 2006), the direct fiber modeling approach demonstrated impressive predictive
159 capabilities. By fitting the model parameters using only one mechanical testing protocol, we were

160 able to reliably predict responses under other loading conditions. We anticipate that this novel
161 direct fiber modeling approach for the sclera will advance our understanding of the biomechanics
162 and mechanobiology of collagen both in healthy and in diseased eyes. Moreover, we expect this
163 approach to be valuable for modeling other soft tissues or tissue-replacement biomaterials.

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166 **2. Methods**

167 This section is organized into three parts: First, biaxial mechanical testing was conducted on
168 healthy sheep posterior pole samples to obtain their stress-strain responses under five loading
169 protocols. Second, the same samples used for mechanical testing were fixed and sectioned, and
170 the sections imaged using PLM. The PLM images were used to build specimen-specific direct
171 fiber models with the fiber orientation data. Each direct fiber model consisted of a fibrous
172 component embedded in a matrix representing the non-collagenous components. Third, each
173 combined fiber and matrix model was used in an inverse fitting optimization process to match the
174 simulated stress-strain behaviors with experimental data acquired under equi-biaxial testing of
175 the same sample. The inverse process produced specimen-specific fiber material properties and
176 pre-stretching strains.

177 Image processing was done using MATLAB v2020 (MathWorks, Natick, MA, USA) and FIJI,
178 which is a distribution of ImageJ (FIJI) (Preibisch et al. 2009; Schindelin et al. 2012). Modeling
179 was done in Abaqus 2020X (Dassault Systemes Simulia Corp., Providence, RI, 171 USA).
180 Customized code and the GIBBON toolbox (Moerman 2018) for MATLAB v2020 (MathWorks,
181 Natick, MA, USA) were used for model pre/post-processing and inverse fitting.

182 **2.1 Biaxial mechanical testing**

183 The experimental procedures were conducted according to a well-established methodology
184 previously described (Amini Khoiy and Amini 2016; Clarin et al. 2023; Khoiy et al. 2018a; Salinas
185 et al. 2020; Thomas et al. 2019). The study was conducted in accordance with the tenets of the
186 Declaration of Helsinki and the statement for the use of animals in ophthalmic and vision research
187 of the Association of Research in Vision and Ophthalmology. Three fresh ovine eyes were
188 obtained from a local slaughterhouse within 24 hours postmortem and immediately transferred to
189 the laboratory at Northeastern University in a cold isotonic phosphate buffer saline (PBS) solution.
190 Upon arrival, two posterior sclera samples measuring 11mm x 11mm were carefully excised from
191 both temporal and superior quadrants, approximately 2 to 3 mm away from the sclera canal
192 (Figure 1 A). For the purpose of optical tracking and tissue deformation/strain analysis, four
193 submillimeter glass markers were affixed to the surface of each sample.

194 The sample was then mounted onto the biaxial testing equipment using fishhooks, with the
195 loading axes aligned with the circumferential and radial directions of the sample (Figure 1 B and
196 C). As demonstrated in our previous studies, to prevent alterations in mechanical response, the
197 samples were continuously submerged in isotonic PBS solution throughout the experiment

198 (Salinas et al. 2019; Salinas et al. 2022). A 0.5 g tare load was initially applied to gently flatten the
199 sample and remove slack without introducing significant pre-stretch. This empirically selected
200 value ensured a consistent, low-strain reference configuration for all samples (Salinas et al. 2019;
201 Salinas et al. 2022). Subsequently, each sample was subjected to biaxial stress control and
202 underwent five distinct loading protocols (Table 1). Each loading protocol consisted of ten 20-
203 second cycles. The first nine cycles served as a preconditioning phase, while data from the last
204 loading cycle were used for subsequent analysis. Based on preliminary tests, a maximum stress
205 level of 150kPa was applied, as it allowed the sample to maintain sample shape without incurring
206 damage. Once the stress-strain data were obtained, the samples were carefully unmounted from
207 the biaxial testing equipment and immersion fixed in 10% formalin for 24 hours. We chose formalin
208 because it has been shown to cause only minimal changes in the shape or size of ocular tissues
209 (Tran et al. 2017).

210 Temporal and superior samples from three eyes produced six samples. Two of the samples
211 (temporal from Eye-1 and superior from Eye-3) were excluded from further analysis because of
212 tissue damage incurred during the experimental procedures. Hence, the rest of the analysis is
213 based on stress-strain responses from four samples. More details regarding these samples can
214 be found in Table 2.

215 At the initial stages of the loading tests, the experimental stress-strain data exhibited relatively
216 high noise and variability compared with more stable later steps. These points were excluded
217 from the inverse fitting process. We applied a moving average smoothing algorithm to reduce the
218 noise in the rest of the stress-strain curves (Chang et al. 2015). For the rest of the analysis, we
219 used the experimental data after smoothing.

220 **2.2 Histology, polarized light microscopy and image post-processing**

221 Following the fixation process, the samples were cryo-sectioned into 20 μm -thick slices
222 (Figure 2). To be able to reconstruct accurately the 3D architecture of the samples, we used a
223 process involving both coronal sections (parallel to the surface of the sample) and sagittal
224 sections (perpendicular to the surface sample). For coronal sections, serial sectioning was
225 performed at the center of the sample, every section was collected starting when there was visible
226 sclera and stopping when the sclera was no longer visible. The number of sections collected
227 varied among the different samples because of their different thicknesses and exact orientation,
228 and the specific quantities are detailed in Table 2. To improve the ability of the reconstructions
229 matching the architecture of each sample, for sagittal sections, two slabs from the edge of the
230 square-shaped sample were obtained. In this way we were able to acquire high resolution coronal

231 data from the core of the tested sample, and high-resolution transverse data at the core edge.
232 Coronal and sagittal sections approximate two orthogonal views of the collagen structure,
233 providing information on the three-dimensional organization of the fibers.

234 All sections were imaged with PLM as described before (Jan et al. 2015; Jan et al. 2017b).
235 Briefly, two polarized filters (Hoya, Tokyo, Japan) were used, one a polarizer and the other an
236 analyzer, to collect images at four filter orientations 45° apart. We have shown that this protocol
237 produces excellent fiber orientation data for ocular tissue samples (Jan et al. 2015). The images
238 were all captured using an Olympus MVX10 microscope (1× magnification setting, 6.84 μm/pixel).

239 PLM images were processed to derive the in-plane collagen orientation at each pixel (in
240 Cartesian coordinates) and a parameter that we referred to as “energy”(Yang et al. 2018). In our
241 PLM analysis, “energy” quantifies the strength of the birefringence signal across different polarizer
242 angles. It is highest in regions where collagen fibers are well-aligned within the imaging plane,
243 and lower in areas with weak signal, such as those lacking collagen or where fibers are oriented
244 out-of-plane. This parameter is used to weight pixel contributions when computing orientation
245 distributions, ensuring that the final histograms reflect meaningful structural information.
246 Following the processing of PLM images, all images obtained from a single sample were
247 sequentially stacked and registered based on the sharp edges (Gogola et al. 2018b). After
248 registration, the original images underwent reprocessing to obtain "corrected" orientation angles
249 that are consistent across all the sections.

250 To focus on a specific area for subsequent construction of the direct fiber model, we selected
251 a square-shaped block positioned at the center of the coronal sections. The dimensions of this
252 selected block were 4.1x4.1mm (Figure 3A). However, due to the irregular shape of the tissue
253 and the folds caused during tissue sectioning, the stack of cropped blocks exhibited variations in
254 thickness (Figure 3C). In order to facilitate fiber tracing for the construction of the direct fiber model,
255 we implemented a linear interpolation algorithm, as described by Akima (Akima 1970; Akima
256 1974). The interpolation algorithm was employed to interpolate angle and energy values at each
257 location in-depth within the stack, thereby converting the inconsistent thicknesses into a uniform
258 value. As a result of this interpolation, the image stack was transformed into a uniform and regular
259 block (Figure 3D). Importantly, this interpolation process did not alter the orientation distribution
260 of the fibers within the stack.

261 To validate the geometry of the direct fiber model, we computed both the coronal and sagittal
262 collagen fiber orientation distributions. For the coronal orientation distribution, we utilized the
263 pixel-level PLM data obtained from the original image stack, taking into account the local "energy"

264 at each pixel. Similarly, for the sagittal orientation distribution, we summed up the PLM data
265 obtained from the cropped block of the two sagittal sections (depicted in Figure 3B). Again, the
266 energy weighting was applied to ensure accurate representation of the collagen fiber orientations
267 in the sagittal plane.

268 **2.3 Direct fiber modeling**

269 **2.3.1 Model construction**

270 We constructed four models (Figure 4A) based on the procedure described previously (Ji et
271 al. 2023a), corresponding to each sample listed in Table 2. Specifically, each modeled domain
272 spanned 4.1 mm × 4.1 mm in-plane (radial and circumferential directions), and the through-
273 thickness dimension varied by sample, ranging from 528 μm to 1168 μm, based on the number of
274 20 μm-thick coronal sections collected (as detailed in Table 2). Fibers were simulated using 3-
275 dimensional linear truss elements (T3D2 in Abaqus). The locations of fibers were defined by
276 sampling orientation values from PLM images at regularly spaced "seed" points (437 μm apart).
277 Straight fibers, 13.68 μm in diameter, were traced at each seed point based on its orientation
278 angle. The process was repeated for each layer, resulting in a stack of 2D layers with
279 interpenetrating fibers. A conflict-resolution algorithm originally developed by Matuschke et al.
280 (Matuschke et al. 2021; Matuschke et al. 2019) was employed to resolve fiber interpenetrations.
281 This algorithm iteratively detected overlapping fiber elements and systematically shifted them
282 apart along their minimal-separation directions until all overlaps were eliminated. Fiber elements
283 were re-meshed to maintain lengths between 82μm and 164μm, while controlling the minimum
284 radius of curvature for smoothness. The amplitudes of the fiber undulations in-depth were
285 adjusted to more accurately represent the distribution of fibers in three dimensions.

286 To account for the natural curvature of the sclera, an additional step was performed to adjust
287 the flat fiber model and match it to the curvature of the eyeball. The external radius of three sheep
288 globes was manually measured, and the average radius was determined to be 14856 μm. The
289 flat model was then projected onto a sphere with an external radius of 14856 μm, effectively
290 introducing curvature to the model. It was important to note that this implied assuming that the
291 sheep eye locally resembled a sphere.(Norman et al. 2010)

292 To assess the similarity between the model and the PLM images, the orientation distribution
293 of the curved model was quantified and compared with the distribution obtained from the PLM
294 images. This comparison aimed to evaluate how well the model captured the observed fiber
295 orientations in the images. The quantification involved counting the occurrences of element

296 orientations within the model, where each element's orientation represented the slope angle in
297 the section plane. This approach accounted for varying element sizes and enabled a proper
298 comparison with the pixel-based measurements obtained from PLM. To evaluate the fitness of
299 the orientation distributions, adjusted R-squared (adjusted R^2) values were employed (Miles
300 2005). The adjusted R^2 values provided a measure of how well the model's orientation distribution
301 fits the distribution observed in the PLM images. The use of adjusted R^2 values allowed for the
302 consideration of the complexity of the model and the number of parameters involved, providing a
303 more robust evaluation metric than a conventional R^2 .

304 In addition to the fiber model, a matrix model was also constructed (Figure 4B). In our model,
305 compatibility was achieved geometrically and through consistent boundary conditions.
306 Specifically, the matrix was designed to match the fiber model in dimensions and curvature, and
307 fiber end-nodes were placed on the surfaces of the matrix. Both components were then subjected
308 to the same displacement boundary conditions during simulation. This setup ensured kinematic
309 compatibility between the fiber and matrix phases throughout the loading process.

310 **2.3.2 Model inverse fitting**

311 **2.3.2.1 Meshing and material properties**

312 Fibers were modeled as a hyperelastic Mooney-Rivlin material (Holzapfel 2001):

$$313 \quad W = C_{10}(I_1 - 3) + C_{01}(I_2 - 3) + \frac{1}{D}(J - 1)^2 \quad (1)$$

314 where W was the strain energy density, C_{10} and C_{01} were the material constants to be determined
315 by inverse fitting. Note that while it is common for both C_{10} and C_{01} to be positive in Mooney-Rivlin
316 models, this is not a requirement. What matters is that the strain energy function remains positive
317 definite (Weizel et al. 2022), i.e., $C_{10} + C_{01} > 0$. I_1 and I_2 were the first and second invariants of
318 the right Cauchy-Green deformation tensor, D determines the compressibility of the material and
319 J was the determinant of the deformation gradient. The matrix was modeled as a neo-Hookean
320 material with a shear modulus of 200 kPa (Coudrillier et al. 2015b; Girard et al. 2009b).

321 The fiber components were modeled using 3-dimensional linear truss elements (T3D2) in
322 Abaqus. The length of the fiber elements ranged from 82 μm to 164 μm , resulting in aspect ratios
323 between 6 and 12. The matrix was meshed using linear eight-noded hybrid hexahedral elements
324 (C3D8H) in Abaqus, where the hybrid formulation introduces a pressure degree of freedom to
325 enforce incompressibility. Accordingly, the parameter D was set to 0, which Abaqus interprets as
326 full incompressibility and handles without numerical instability (Corp 2014). The element size for

327 the matrix varied among the samples due to differences in sample thickness, with 4 to 6 elements
328 spanning the shell thickness. To ensure model accuracy, a mesh refinement study was conducted.
329 The fiber model's mesh density was doubled, while the matrix model had its mesh density doubled
330 in both in-plane directions and the thickness direction. The study's findings indicated that altering
331 the mesh density had a negligible impact on stress predictions, with maximum stress values
332 changing by less than 1%. Based on these findings, the chosen mesh density was deemed
333 sufficient to ensure numerical accuracy in the obtained results.

334 **2.3.2.2 Interactions**

335 Fiber-fiber interactions were simulated in the following two ways. First, the interactions were
336 considered by preventing fiber interpenetrations using Abaqus' general contact with no friction.
337 Second, to enhance the interweaving effect of fibers and prevent them from sliding apart during
338 stretching, a method involving the constraint of nodes was employed. Approximately 10% of the
339 nodes, primarily located in the outer surface or boundary of the model, were selected. These
340 nodes were connected to their closest neighboring nodes, and their relative motion in the Z
341 direction was constrained to zero using linear constraint equations in Abaqus. By applying these
342 constraints, the free ends of the fibers were better controlled, resulting in a more stable and
343 realistic model.

344 Fiber-matrix interactions were ignored, as is usual in biomechanical models of the eyes (Grytz
345 et al. 2014; Grytz and Meschke 2009; Petsche and Pinsky 2013). We recognize that in actual
346 tissues there are interactions between the fibers and other fibers and the fibers and the matrix.
347 Here it is important to point out that many conventional approaches to model soft tissues that use
348 statistical and homogenization approaches disregard both interactions (Grytz et al. 2014; Grytz
349 and Meschke 2009; Petsche and Pinsky 2013). Although our direct fiber modeling approach does
350 not yet include detailed modeling of fiber-matrix interactions, it does account for fiber-fiber
351 interactions and is therefore a step in the right direction.

352 **2.3.2.3 Finite element analysis procedure**

353 The fiber–matrix assembly was subjected to a quasi-static biaxial stretching process to
354 replicate the experimental stress–strain data described in Section 2.1. Because mechanical
355 interaction between the fiber and matrix domains was not modeled, the two components were
356 treated as mechanically independent and solved separately using different finite element
357 procedures. The matrix was modeled using the Abaqus standard implicit solver with C3D8H
358 elements, suitable for simulating nearly incompressible soft tissue behavior. The fibers,

359 represented as T3D2 truss elements, were simulated using the explicit dynamic solver, which is
360 more robust for resolving complex fiber–fiber contact interactions. The resulting stresses σ in the
361 radial and circumferential directions were a combination of matrix and fiber contributions, with the
362 matrix stress weighted by the fiber volume fraction (VF) as shown in Equation (2):

$$363 \quad \sigma = (1 - VF)\sigma_{\text{matrix}} + VF\sigma_{\text{fibers}} \quad (2)$$

364 To ensure efficient dynamic analysis, mass scaling was implemented, allowing for a stable
365 time increment of $1e-5$. The time step was selected to satisfy quasi-static loading conditions,
366 whereas different values may be required to capture high-frequency dynamic responses. To
367 confirm that the simulation remained quasi-static and that the mass scaling and time step did not
368 introduce significant dynamic effects, the kinetic-to-internal energy ratio was monitored and
369 remained below 5% (Supplemental data - Figure S5), a commonly accepted threshold for
370 negligible inertial effects (Corp 2014).

371 **2.3.2.4 Boundary conditions and inverse fitting procedure**

372 In the biaxial stretching process of the fiber-matrix assembly (Figure 4C), a two-step approach
373 was employed to simulate the experimental conditions. In the first step of the biaxial stretch
374 simulation, radial and circumferential pre-stretching strains were applied to the model boundaries
375 by prescribing displacements corresponding to those strain values as shown in Figure 4C. These
376 pre-stretches represent the deformations caused by the 0.5g tare load in the physical experiments.
377 Because these experimental strains were not directly measured, the radial and circumferential
378 pre-stretch values were treated as unknowns and estimated through the inverse modeling
379 procedure. Specifically, the inverse fitting algorithm adjusted these displacement boundary
380 conditions (i.e., pre-stretching strains) in combination with the fiber material parameters to
381 minimize the difference between the simulated and experimental stress-strain responses under
382 equibiaxial loading. In the second step, the model was further stretched, and the same strains
383 observed in the experiment were assigned as the displacement boundary condition to the fiber-
384 matrix model. In this step, the model stress and strain values were recorded, starting from 0. This
385 approach was consistent with the experimental setup, where the stress and strain resulting from
386 the tare load were not included in the reported experimental stress-strain data. By applying this
387 two-step biaxial stretch with the appropriate displacement boundary conditions, the model aimed
388 to replicate the mechanical behavior observed in the experiment and allow for a comparison
389 between the model's stress-strain behaviors and the experimental data.

390 During the inverse modeling procedure, there were four parameters to determine. The first
391 two parameters were the pre-stretching strains along the radial and circumferential directions in
392 the first step of the biaxial stretch. Since the resultant strains caused by the tare load were not
393 characterized in the experiment, these values were unknown and needed to be determined
394 through the inverse modeling process. The second two parameters were the material properties
395 of the fibers (C_{10} and C_{01}).

396 In the inverse fitting procedure, two optimizations were performed to determine the optimal
397 parameters and match the stress-strain data of the five loading protocols. In the first optimization,
398 all four parameters (pre-stretching strains and fiber material properties) were optimized to match
399 the stress-strain data obtained using loading protocol 1:1. Further discussion on the selection of
400 loading protocol 1:1 can be found in the Discussion section. A simplex search method was used
401 for this optimization (Lagarias et al. 1998). The algorithm aimed to find the values of the
402 parameters that minimized the residual sum of squares (RSS) between the simulated and
403 experimental stress-strain curves. The optimization process continued until the adjusted R^2 value
404 between the curves exceeded 0.9, indicating a good fit between the model and experimental data.
405 This optimization allowed for the determination of the optimal pre-stretching strains and fiber
406 material properties (C_{10} and C_{01}).

407 In the second optimization, only the two pre-stretching strains were optimized, using the
408 derived fiber material properties from the first optimization. This optimization aimed to match the
409 stress-strain data obtained from the remaining four loading protocols. The same search method
410 as before was employed, and the optimization was concluded when the adjusted R^2 value
411 between the stress-strain curves exceeded 0.9 in both loading directions.

412 By performing these two optimizations, the model was able to find the optimal parameters that
413 resulted in stress-strain curves closely aligned with the experimental data for all the five loading
414 protocols.

415 Fiber material properties were obtained by matching experimental data from the equi-biaxial,
416 1:1, loading protocol. The natural question thus arises: to what extent would the results differ if
417 the material properties were obtained by matching experimental data from other loading protocols?
418 To address this, we selected one Sample, which had the best fitting accuracy using fiber material
419 properties derived from the 1:1 loading protocol. We then independently derived the fiber material
420 properties by fitting the experimental data from the 1:0.75, 0.75:1, and 1:0.5 loading protocols.
421 The fitting process began with random initial values for the material properties, which were

422 iteratively optimized to match the experimental data. Importantly, each fitting process was
423 conducted independently, without reference to results from other loading protocols.

424 To evaluate the physical significance and consistency of the derived fiber material properties,
425 we focused on two key mechanical properties: the fiber elastic modulus and the shear bulk
426 modulus. These properties are critical for understanding the mechanical behavior of the fibers
427 under different loading conditions. The elastic modulus characterizes the stiffness of the fibers,
428 while the shear bulk modulus reflects their resistance to shear deformation. Calculating these
429 properties allowed us to assess whether the derived material properties were physically
430 meaningful and consistent across different loading protocols.

431 The fiber elastic modulus was estimated by simulating uniaxial stretch of a single straight fiber
432 using the optimal C10 and C01 values of the hyperelastic Mooney-Rivlin material. This approach
433 provided a direct measure of the fiber's stiffness under tensile loading. The shear bulk modulus,
434 on the other hand, was derived from the fitted material properties by analyzing the fiber's response
435 to shear deformation. Both properties were calculated for each set of derived material properties
436 obtained from the different loading protocols.

437 **Preliminary analysis:** During the preliminary analysis of the inverse fitting procedure using
438 Sample #1, it was observed that the experimental data from loading protocol 0.5:1 exhibited
439 negative strains along one loading direction. This suggested the presence of anisotropic
440 mechanical properties, with one loading direction being much softer than the other direction.
441 When assigning negative strains to the direct fiber model, it led to instability in the model. This
442 was problematic because the direct fiber model is capable of accurately representing fiber tension,
443 but not as well longitudinal fiber compression and buckling. As a result, all the loading protocols
444 that showed negative strains were excluded from the analysis. Based on the mechanical testing
445 results, the loading protocol 0.5:1 of all the samples presented negative strains (Supplemental
446 data - Figure S1). Therefore, the results related to loading protocol 0.5:1 were not reported in this
447 study. The focus was placed on the remaining loading protocols that did not exhibit negative
448 strains.

449

450

451 **3. Results**

452 Figure 5 presents the fiber orientation distributions of the direct fiber model and corresponding
453 PLM images for the four samples. The agreement between the model and PLM images was
454 observed in both the coronal and sagittal planes. The alignment of these orientation curves
455 yielded adjusted R^2 values exceeding 0.89 in all cases, indicating a strong match between the
456 model and experimental data.

457 Figure 6 shows fiber displacements and stresses at full stretch of an example sample (Sample
458 #4). Results for the other three samples are provided in the Supplementary Material (Figures S2–
459 S4). The visualization revealed heterogeneous behaviors of the fibers, with varying deformations
460 and stress distributions at the microscale. The model effectively captured the non-uniform
461 response of the fibers under applied loading conditions, highlighting the intricate mechanical
462 behavior within the tissue.

463 Isometric views of the Sample #4 direct fiber model during loading protocol 1:1 are shown in
464 Figure 7. The visualization uses color to represent the maximum principal stress (left column) and
465 displacement magnitudes (right column). Initially, the model showed curvature, which gradually
466 flattened during stretching. As expected, more fibers experienced higher stress as stretching was
467 applied, in a process of recruitment.

468 Stress-strain curves for all loading protocols, predicted by the optimized models using material
469 parameters derived from the equi-biaxial loading protocol, are shown in Figure 8. These curves
470 demonstrate decent agreement with experimental data in both the radial and circumferential
471 directions for most loading protocol cases with R^2 values exceeded 0.85. Note that high values
472 for the adjusted R^2 , indicating a robust fit between the model predictions and experimental
473 observations. The derived parameters that achieved this high level of agreement are provided
474 in Table 3.

475 Figure 9 presents the fiber orientation distributions of the direct fiber model and the model's
476 mechanical anisotropy at maximum strain state. The findings suggest that the stiffness along each
477 direction is approximately proportional to the amount of fibers in the loading direction at the
478 maximum strain state.

479 The derived fiber elastic and shear modulus demonstrated remarkable consistency across the
480 results obtained from fitting different loading protocols of Sample #4, as shown in Figure 10. This

481 consistency underscores the robustness of our method and confirms that the derived material
482 properties are largely independent of the choice of loading protocol used for fitting.

483

484 **4. Discussion**

485 Our goal was to continue advancing direct fiber models of soft tissues. Specifically, we aimed
486 to conduct a more challenging test of the ability of the direct fiber modeling approach to capture
487 sclera microstructure and anisotropic mechanics. We developed direct fiber models to simulate
488 four sclera samples, incorporating specimen-specific 3D fiber architecture. Subsequently, we
489 conducted an inverse fitting study and matched the models with specimen-specific anisotropic
490 stress-strain behaviors from biaxial testing of the same samples. The study yielded two main
491 findings. First, direct fiber model reconstruction can capture the 3D collagen structure of multiple
492 sclera samples from different quadrants and eyes. Second, the macroscopic mechanical
493 properties of the models matched with the experimental stress-strain data of the same sample
494 obtained under various anisotropic loading conditions. Notably, this was achieved by having fit
495 the models only to the equi-biaxial experiment. The derived material properties were then also
496 shown to be appropriate for the other loading conditions. This indicates that the direct fiber models
497 inherently incorporated the anisotropy of tissue mechanical behaviors within their fiber structure,
498 thereby eliminating the necessity for separate optimization with different loading conditions. Below
499 we discuss the findings in detail.

500 **Finding 1. Direct fiber model reconstruction can capture the 3D collagen structure of** 501 **multiple sclera samples from different quadrants and eyes.**

502 In our previous study, the primary focus was on introducing the direct fiber modeling
503 methodology, which involved utilizing experimental data from different samples and species
504 obtained for other research purposes (Ji et al. 2023a). In this study we aimed to reconstruct and
505 match the behavior of the same sample. To maintain the broad applicability, we chose to construct
506 models of multiple sclera samples from different quadrants and eyes. This decision was driven
507 by the fact that the collagen fiber architecture of the sclera varies both spatially between locations
508 and between different eyes (Gogola et al. 2018b). Furthermore, it is known that the mechanical
509 behavior of the sclera exhibits varying degrees of anisotropy across different quadrants
510 (Coudrillier et al. 2013; Hua et al. 2022). By employing samples from both superior and temporal
511 quadrants of multiple eyes, our study successfully demonstrated the robustness of the direct fiber
512 model reconstruction approach in capturing the varying fiber structure. It is important to
513 emphasize that our models were specimen-specific, meaning that the fiber structure was
514 individually constructed based on each sample, and the model's behaviors were optimized using
515 experimental data obtained from the same sample. Interestingly, the fiber orientation distributions
516 obtained from the PLM data (red curves in Figure 5) are smoother than those derived from the

517 direct fiber models (green curves). This is because the PLM-based orientations sample actual
518 tissues with a very large number of fibers, whose orientations are captured at pixel resolution and
519 are subject to noise, whereas the model orientations are obtained from a much smaller set of fiber
520 segments whose orientations are known with very high precision. Increasing the number of fiber
521 segments would lead to smoother distributions, but at higher computational cost.

522 The morphological parameters used to construct the fiber models—specifically the fiber
523 spacing (seeding density) and diameter—were selected based on a series of preliminary tests to
524 achieve an appropriate balance between capturing collagen architecture, maintaining realistic
525 tissue thickness, and ensuring computational feasibility. The seeding density directly influenced
526 both the computational cost and the ability of the model to represent the underlying collagen
527 architecture. A denser seeding of fibers provided a more accurate match to the orientation
528 distributions derived from PLM images (as shown in Figure 5) but substantially increased the
529 number of fibers and computational complexity. In contrast, sparse seeding reduced
530 computational burden but risked losing key structural features. Fiber diameter, on the other hand,
531 primarily impacted the model’s thickness after resolving fiber–fiber interpenetrations. Larger fiber
532 diameters necessitated greater shifts to resolve interpenetrations, which cumulatively led to an
533 inflated tissue thickness relative to the original PLM-imaged block. Conversely, selecting a very
534 small diameter reduced this effect but yielded unrealistically low fiber volume fractions and
535 insufficient thickness. While PLM images can, in principle, resolve fiber dimensions, estimating
536 diameter directly from them would require substantially denser sampling, further increasing
537 computational demands. The parameters chosen in this study therefore reflect a deliberate
538 compromise that enables realistic geometry, structural fidelity, and computational efficiency.

539 **Finding 2: The macroscopic mechanical properties of the models matched with the**
540 **experimental stress-strain data of the same sample obtained under various anisotropic**
541 **loading conditions.**

542 The importance of this finding is probably best understood when seen in the context of the
543 conventional way that constitutive models of soft tissues are developed to incorporate fiber
544 microstructure and nonlinear behavior. Prior studies have reported that a single equi-biaxial
545 loading experiment is insufficient for predicting other loading responses (He and Lu 2022; Thomas
546 et al. 2019; Zhang et al. 2019). To achieve the necessary predictive accuracy for other loading
547 cases with uneven loading ratios, the equi-biaxial loading case must be supplemented with two
548 additional loading conditions. Thus, for instance, the set of experimental tests used in this work

549 was developed and proven adequate to capture tissue properties using five experimental
550 conditions, three for model fitting and two for fit testing. This requirement indicates that the
551 conventional continuum models fail to capture at least part of the intrinsic anisotropy of the
552 microstructure, which must then be supplemented by more continuum parameters that require
553 other loading tests to fit. This complexity arises in large part because the nonlinear behavior of
554 collagenous tissues depends on fiber characteristics, including undulations across multiple
555 scales—with crimp representing a well-known small-scale periodic undulation observed in
556 collagen fibers (Islam et al. 2024; Jan and Sigal 2018; Lee et al. 2024) —as well as fiber-fiber
557 interactions (Wang et al. 2020). The conventional approach of capturing fiber architecture in an
558 element-by-element basis using one or more statistical distributions of fiber orientations (and
559 potentially other characteristics) has only a limited ability to describe the complexity of the tissue
560 microarchitecture. While techniques have been successfully implemented to account for fiber
561 crimp (Foong et al. 2023; Grytz et al. 2014; Grytz and Meschke 2009; Hill et al. 2012), the
562 approach is particularly limited in accounting for fiber-fiber interactions.

563 It is important to acknowledge that our direct fiber reconstruction approach still results in
564 an approximation of the actual fibrous structure of the sclera (more on this later in the subsection
565 on Limitations). Nevertheless, we argue that it represents a step forward for specimen-specific
566 modeling.

567 **Interpretation of the derived fiber material properties**

568 The “uniqueness” of so-called optimal parameters is a common concern in inverse fitting
569 techniques (Girard et al. 2009a; Zhou et al. 2019). To evaluate this, we conducted the inverse
570 fitting process using four different sets of starting parameters. While the derived fiber material
571 properties (C10 and C01) exhibited significant variation across the four samples, the estimated
572 fiber elastic modulus ranged from 0.47 GPa to 3.22 GPa for Samples #2, #3 and #4 (see Table
573 3). This range aligns with experimentally reported values in the literature: 0.2 GPa to 7 GPa for
574 bovine Achilles tendon fibers (van der Rijt et al. 2006; Yang et al. 2007), and 5 GPa to 11.5 GPa
575 in rat tail (Wenger et al. 2007). For Sample #1, the estimated fiber modulus was 23.97GPa , which
576 falls outside the experimentally reported ranges. This could be attributed to two potential factors.
577 First, the load may have been primarily borne by a small portion of the fibers, leading to an
578 overestimation of the fiber material properties. Second, the fiber volume fraction in our models
579 (approximately 15%) may not accurately represent the actual tissue composition. Experimental
580 data on scleral fiber volume fraction spans a wide range, from 18% to 93% (Boote et al. 2016;
581 Cone-Kimball et al. 2013; Edwards and Prausnitz 1998; Quigley et al. 1991). While there are

582 substantial variations between individual sclera, it is likely that a substantial contribution to the
583 wide range is due to differences between methodologies, as we have discussed elsewhere (Wang
584 et al. 2020). The derived fiber material properties are closely tied to the fiber volume fraction in
585 the reconstructed models. However, reconstructing models with higher fiber density is challenging
586 due to the increasing difficulty in tracing fibers and replicating their turns and undulations as the
587 density increases (Yang et al. 2021). Further research is needed to comprehensively understand
588 how variations in fiber volume fraction influence the behavior and derived material properties of
589 the direct fiber model.

590 Another concern is whether the derived material properties and pre-stretching strains
591 could lead to an overly favorable match between the models and experimental data across all
592 loading protocols. To address this, we conducted a specific analysis using the fiber structure of
593 Sample #1 and attempted to match the experimental data of Sample #2. While the model was
594 able to capture the general stress-strain behavior of Sample #2 under the 1:1, 1:0.75, and 0.75:1
595 loading protocols using Sample #1's fiber architecture, it visibly failed to do so under the 1:0.5
596 protocol. This contrast highlights the importance of sample-specific fiber structure for accurate
597 mechanical prediction (Figure 11). It is noted that the fitting accuracy for the 1:0.5 loading protocol
598 in Sample #2 is not excellent, as evidenced by the single fitting result in Figure 8, which yielded
599 an adjusted R^2 with 0.755 in the circumferential direction and 0.580 in the radial direction.
600 However, the accuracy decreased to below 0.5 in the cross-fitting analysis. These results indicate
601 that the model's applicability is not universal but depends on accurately matching the orientation
602 distribution of the specific sample being studied. The relatively poor fitting accuracy for the 1:0.5
603 loading protocol further highlights the importance of sample-specific fiber orientation in replicating
604 experimental data.

605 **Strengths of direct fiber models**

606 The models developed in this study have several strengths, some of which we have discussed
607 above or elsewhere (Ji et al. 2023a). Briefly, the direct fiber models incorporate fiber interweaving
608 and the resulting fiber-fiber interactions that play an important role in determining the structural
609 stiffness of the sclera (Wang et al. 2020). The models also include collagen fibers of the sclera
610 that are long and continuous. Thus, they can transfer forces over a long distance (Boote et al.
611 2020; Voorhees et al. 2018), that may also play important roles in tissue mechanics (Grytz et al.
612 2020; Huang et al. 2017; Islam et al. 2024; Lanir 2017). This is in contrast to the conventional
613 continuum mechanics approaches where fibers exist only within a given element, disrupting load
614 transmission at the element boundaries (He et al. 2024). Furthermore, the direct fiber models in

615 this work incorporate three-dimensional specimen-specific collagen architecture, whereas
616 previous fiber-aware models tend to overlook or simplify the in-depth orientations of the collagen
617 fibers (Coudrillier et al. 2013; Voorhees et al. 2017; Zhang et al. 2015). Collagen fiber variations
618 in-depth of the tissue are crucial in determining the sclera's load-bearing capacity, particularly in
619 bearing shear stresses, and may have clinical implications (Danford et al. 2013; Ji et al. 2023b).

620 The observed variability in sample thickness (Table 2) plays a structurally mediated role
621 in influencing biomechanical behavior. In our experimental setup, stress was computed by
622 normalizing force with the cross-sectional area, including measured thickness, ensuring that the
623 reported stress-strain relationships reflected intrinsic material behavior rather than geometry-
624 driven effects. However, the impact of thickness goes beyond geometry in our direct fiber
625 modeling framework. Thicker samples contain a larger number of PLM-derived coronal sections,
626 which leads to the inclusion of a broader range of fiber orientations through the depth of the tissue.
627 This increased through-thickness heterogeneity in fiber directions can contribute to a more
628 isotropic representation of collagen architecture. This phenomenon is evident in our results.
629 Samples with greater thickness—specifically, Sample 2 (1168 μm) and Sample 4 (838 μm)—
630 exhibited relatively isotropic mechanical and structural behavior (Figure 9), while the thinner
631 samples, Sample 1 (888 μm) and Sample 3 (796 μm), showed more pronounced anisotropy.
632 These observations suggest that sample thickness may influence apparent anisotropy through its
633 role in shaping the depth-wise distribution of collagen fibers, which our modeling approach is
634 specifically designed to capture.

635 In comparison to our previous studies, we implemented two improvements to the process of
636 building direct fiber model structures. Firstly, we incorporated the natural curvature of the tissue
637 into the model, enabling a more accurate representation of the physiological shape of the sclera.
638 This enhancement ensured that the model mimicked more closely the three-dimensional
639 geometry of the sclera, improving its fidelity. Secondly, we introduced a constraint function in
640 Abaqus to deal with fibers that were not sufficiently constrained. This helped avoid, for example,
641 cases where a fiber could be “pulled out” of the model. This function served to tightly constrain
642 the interconnections between fibers. As a result, the model not only exhibited increased stability,
643 but also better matched the actual interweaving and cohesion of collagen fibers in the sclera. This
644 optimization helped overcome the challenge of inaccuracies caused by floating fibers and
645 enhances the model's reliability and validity in representing the real tissue structure. We suspect
646 that this constraint would not be necessary if we had modeled in more detail the fiber-fiber and
647 fiber-matrix interactions. The simplifications we assumed on these interactions prevented

648 crossing and interpenetration, but nothing else. This allowed fibers much freedom to displace and
649 slide. The roles of proteoglycans on this aspect of sclera behavior is important and still not well
650 understood (Hatami-Marbini and Pachenari 2020; Murienne et al. 2015). It is also possible that
651 our modeling of somewhat thick fiber bundles instead of small scale fibers may not have captured
652 the full extent of the fiber entanglement that limits fiber sliding. Further work is necessary to
653 characterize these aspects of the sclera and how to account for them accurately in models. Herein
654 we just want to comment that we acknowledge that our treating of fiber-fiber and fiber-matrix
655 interactions, while limited, is still more comprehensive than that in conventional continuum models
656 where these are generally not only ignored but cannot be incorporated without cumbersome
657 kludges. Our direct fiber modeling approach highlights the assumptions on the interactions and
658 provides a platform for improving their modeling.

659 **Limitations**

660 The first limitation we would like to comment on was observed in the matching of stress-strain
661 curves between the model and experimental data. While the majority of loading conditions
662 exhibited strong agreement with adjusted R^2 values exceeding 0.9, there were instances,
663 particularly in the loading protocol 1:0.5, where the model's fit was comparatively low. We do not
664 know the cause, but this discrepancy may result from the sequential application of five different
665 loading protocols during the testing process. After multiple rounds of testing at levels near the
666 tissue limits, the tissue may have been affected. As a result, the model's behavior appeared to be
667 stiffer than the corresponding experimental data in these cases. Future studies should either
668 randomize the loading protocols or explicitly study the effects of protocol order. Despite this
669 limitation, it is important to emphasize that the direct fiber modeling technique demonstrated
670 robustness and effectiveness in capturing the overall mechanical behaviors of the sclera, as
671 evidenced by the strong agreement between the model and experimental data in the majority of
672 cases. The observed discrepancies in certain loading conditions provide valuable insights for
673 future refinements of the model and highlight the need for further investigation into the dynamic
674 changes in tissue properties during sequential loading.

675 Second, we encountered difficulties in accurately matching the loading protocol 0.5:1 due to
676 the model's inability to perform well under compression. This limitation indicated that further
677 optimization of the approach is required to enable stable simulations under compression. Again,
678 we suspect that this is largely due to the simplifications in fiber-fiber interactions considered
679 (Hatami-Marbini and Pachenari 2020; Murienne et al. 2015). Future efforts should be focused on
680 addressing this issue to enhance the applicability of the direct fiber modeling approach.

681 Third, we used sheep eyes, and thus the question arises as to how well this would work with
682 other tissues, or with other species. Considering the robustness of the approach in capturing the
683 complex fiber structure and anisotropic mechanical behaviors of sheep sclera, we anticipated that
684 the direct fiber modeling technique can serve as a valid and effective tool for investigating scleral
685 biomechanics in other species and potentially in other collagen-based tissues as well. Tissues
686 that approximate transverse isotropy, like the sclera, may fare better when imaged under PLM.
687 More 3D isotropic tissues may require other imaging techniques. Future studies can extend the
688 application of this approach to validate its broader utility and ensure its broad application across
689 various tissues.

690 Fourth, only four samples were analyzed in this study. While the sample size was limited, the
691 objective of the study was to validate the direct fiber modeling method using well-controlled,
692 specimen-specific data, rather than to establish tissue characteristics across the population or
693 other generalizations. These would likely require more samples. Whilst consistent performance
694 across samples and loading protocols indicates method robustness, it should not be assumed
695 that the method works equally well for other tissues.

696 Fifth, the direct fiber models, while consistent with experimentally measured fiber orientation
697 distribution, are still approximations of the actual tissue. Several aspects were simplified or
698 ignored in the modeling process, which may introduce discrepancies between the models and the
699 real tissue. One such simplification was the assumption of uniform fiber or bundle diameters within
700 the models. In reality, collagen fibers in the sclera exhibit variations in diameter (Gogola et al.
701 2018a; Jan et al. 2018; Komai and Ushiki 1991; Paidimarri et al. 2025). Additionally, sub-fiber
702 level features were not explicitly included in the models, such as fiber crimp (Grytz et al. 2014;
703 Lee et al. 2022). Future work would benefit from incorporating more detailed and realistic
704 microstructural features, where the models can provide a more accurate representation of the
705 scleral tissue and enhance our understanding of its mechanical behavior.

706 Sixth, the matrix mechanical properties were kept constant at literature values and not optimized
707 iteratively like the fiber properties. This simplification was made for simplicity. Although the matrix
708 properties could potentially affect fiber load-bearing and parameter fitting, their impact is
709 considered minor. The fibers, being the primary load-bearing component, exhibit significantly
710 greater stiffness compared to the matrix (Coudrillier et al. 2015a; Girard et al. 2009a). Analysis of
711 our model indicated that the matrix bears only 4%–6% of the total reaction forces at the maximum
712 strain. Therefore, we believe that the fiber properties dominate the model's behavior, while the
713 matrix stiffness has minimal influence on the derived parameters. In future work, it would be

714 worthwhile to explore the role of matrix properties and consider more detailed matrix-fiber
715 interactions for further refinement.

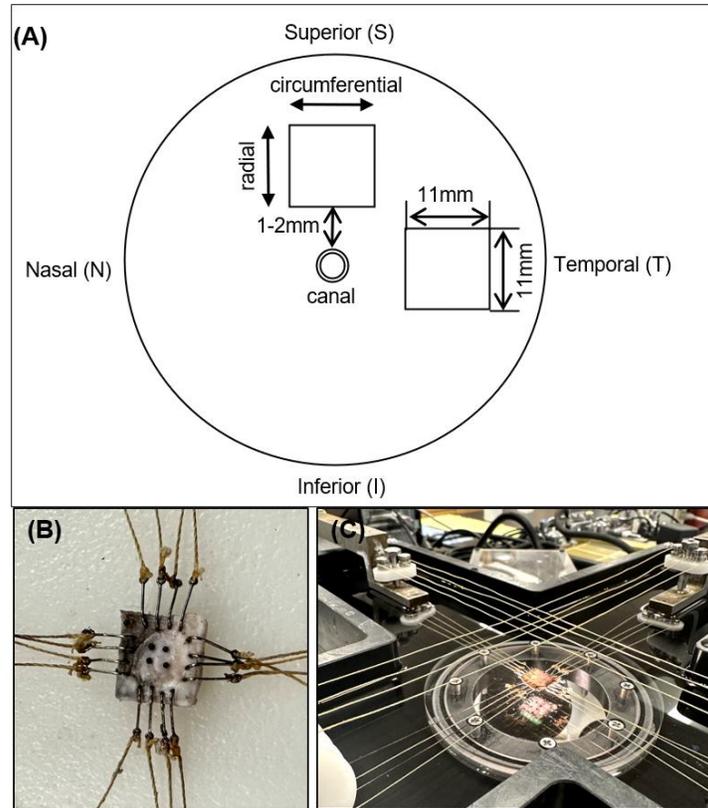
716 Seventh, we ignored fiber-matrix interactions that can be quite complex and have important
717 effects on tissue behavior (Hatami-Marbini and Pachenari 2020; Muriene et al. 2016). While this
718 is routinely done in continuum modeling of soft tissues, it is still a potentially important oversight.
719 Interesting ideas have been proposed on how to deal with fiber-matrix interactions (Kakaletsis et
720 al. 2023), but further research is needed to ensure that they do not overly constrain fiber behavior.

721 Eighth, in the current direct fiber model, we used T3D2 truss elements to represent collagen
722 fibers. These elements support only axial loading (tension and compression) and neglect bending
723 and torsional resistance. This modeling decision is justified by the slender geometry and tensile-
724 dominated mechanics of collagen fibers, particularly under the planar biaxial loading conditions
725 studied here. However, we acknowledge that excluding bending stiffness may limit the model's
726 applicability in scenarios involving significant local curvature, compressive loads, or mechanical
727 constraints that force fibers to reorient or buckle. Importantly, this limitation is not unique to our
728 model: most statistical and homogenization-based approaches similarly neglect compression and
729 buckling, often in even more simplified ways. Future extensions of our framework could
730 incorporate beam elements or hybrid truss–beam formulations to enable more comprehensive
731 modeling, especially for tissues subjected to complex geometries or loading environments.

732 It would be interesting in future work to compare between flat and curved models with identical
733 microstructural inputs to elucidate the role of curvature in fiber recruitment, load distribution, and
734 macroscopic anisotropy.

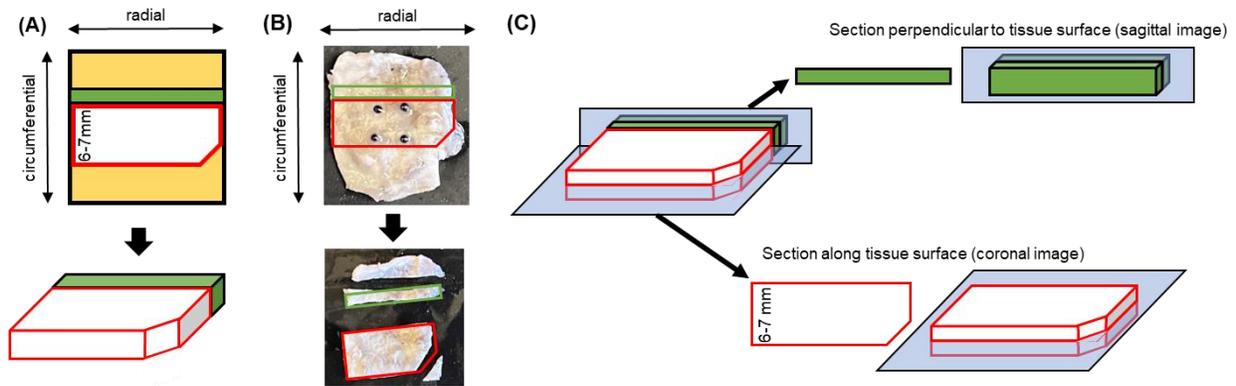
735 In conclusion, we performed a comprehensive and challenging test of the direct fiber modeling
736 approach through the simulation of multiple sheep posterior sclera samples. We characterized
737 the macroscopic and anisotropic stress-strain behaviors of the samples through biaxial
738 mechanical testing. Then direct fiber models were generated based on the microstructural
739 architecture of each sample. An inverse fitting process was employed to simulate biaxial
740 stretching conditions, enabling the determination of optimal pre-stretching strains and fiber
741 material properties. Our findings demonstrated the efficacy of the direct fiber modeling approach
742 in simulating the scleral microarchitecture, capturing critical fiber characteristics, and accurately
743 describing its anisotropic macroscale mechanical behaviors. Moreover, we highlighted the
744 capability of the direct fiber models to inherently incorporate the anisotropy of tissue mechanical

745 behaviors within their fiber structure. Overall, the direct fiber modeling approach proved to be a
746 robust and effective tool for characterizing the biomechanics of sclera and other fibrous soft tissue.



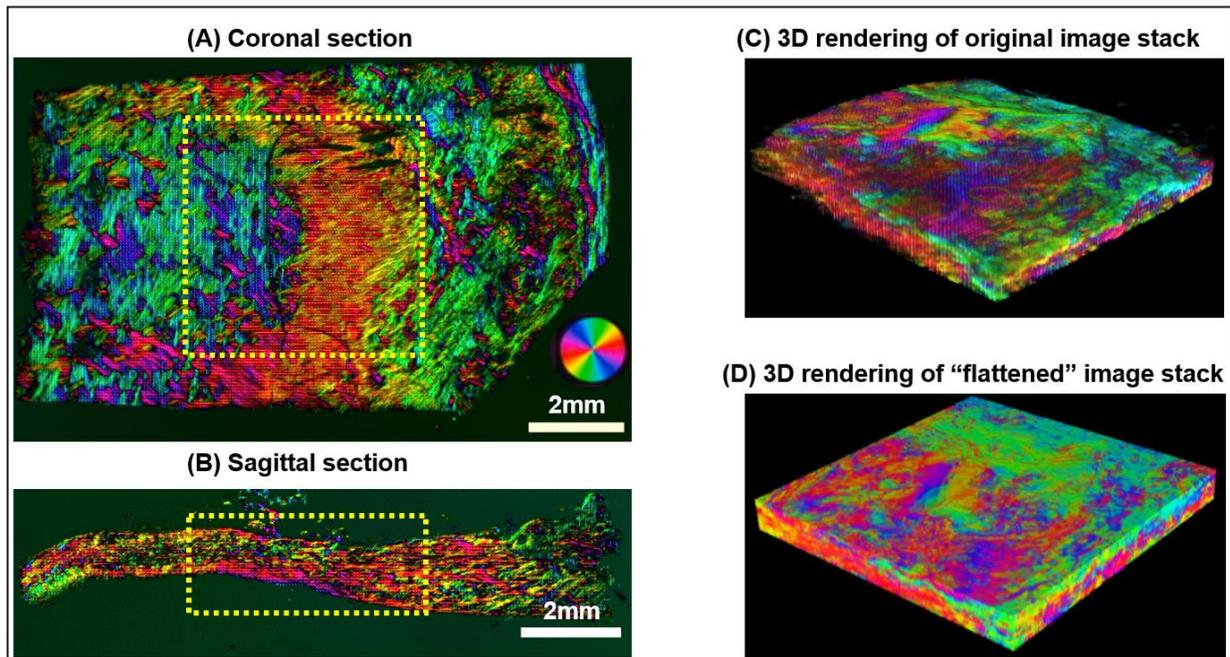
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748 **Figure 1.** Key aspects of the experimental setup and procedures for the biaxial testing. (A)
 749 Schematic diagram showcasing the posterior pole of the eye, highlighting the locations of the two
 750 samples (depicted as yellow boxes) that were obtained specifically for the biaxial testing. Each
 751 sample measured 11 mm × 11 mm in the in-plane dimensions. The corresponding sample
 752 thicknesses are listed in Table 2. (B) An example image showing a sample with hooks attached,
 753 ready for biaxial testing. The hooks ensured proper mounting and fixation of the sample during
 754 the experimental procedures. (C) The sample was mounted on a custom-built biaxial mechanical
 755 testing system, while continuously submerged in isotonic PBS solution. The loading axes of the
 756 system were aligned with the circumferential and radial directions of the sample, ensuring precise
 757 application of stress in the desired directions. The sample was then subjected to the five loading
 758 protocols. Note that the crossed strings, as depicted at the lower right of the sample, were
 759 corrected prior to conducting the experiment.



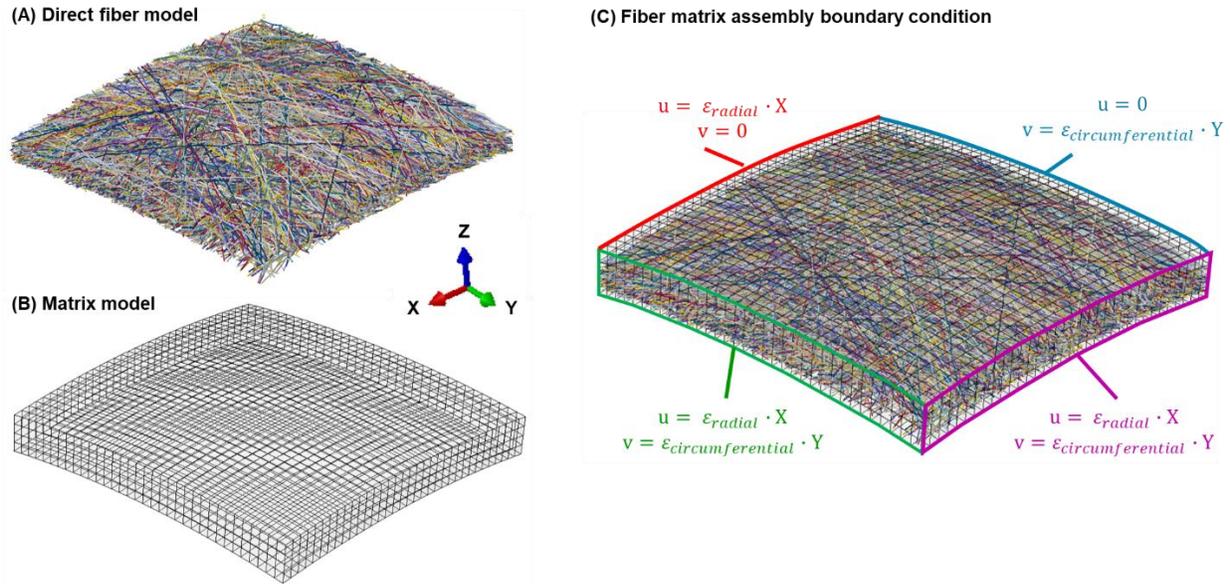
760

761 **Figure 2.** Process of sectioning a sclera sample. **(A)** Sclera sample after biaxial testing was
 762 processed for sectioning (top: 2D view; bottom: 3D view of the region for sectioning). A notch at
 763 the corner of the sample was used to indicate the tissue directions. Since it was not feasible to
 764 section the same piece of tissue both coronally and sagittally, a sample (depicted as a white block,
 765 the shorter length was 6-7mm) was obtained from the center of the tissue. Additionally, another
 766 sample was acquired from the adjacent tissue next to the white block (shown as a green block).
 767 **(B)** An example image of the sclera tissue with the two samples dissected. Prior to dissection, the
 768 fiberglass markers attached to the tissue were carefully removed. **(C, top)** The green block was
 769 sectioned sagittally, resulting in sagittal sections of the tissue. **(C, bottom)** The white block was
 770 coronally sectioned, allowing for the acquisition of serial sections without any loss. The blue
 771 surface depicted in the image represented the plane of sectioning.



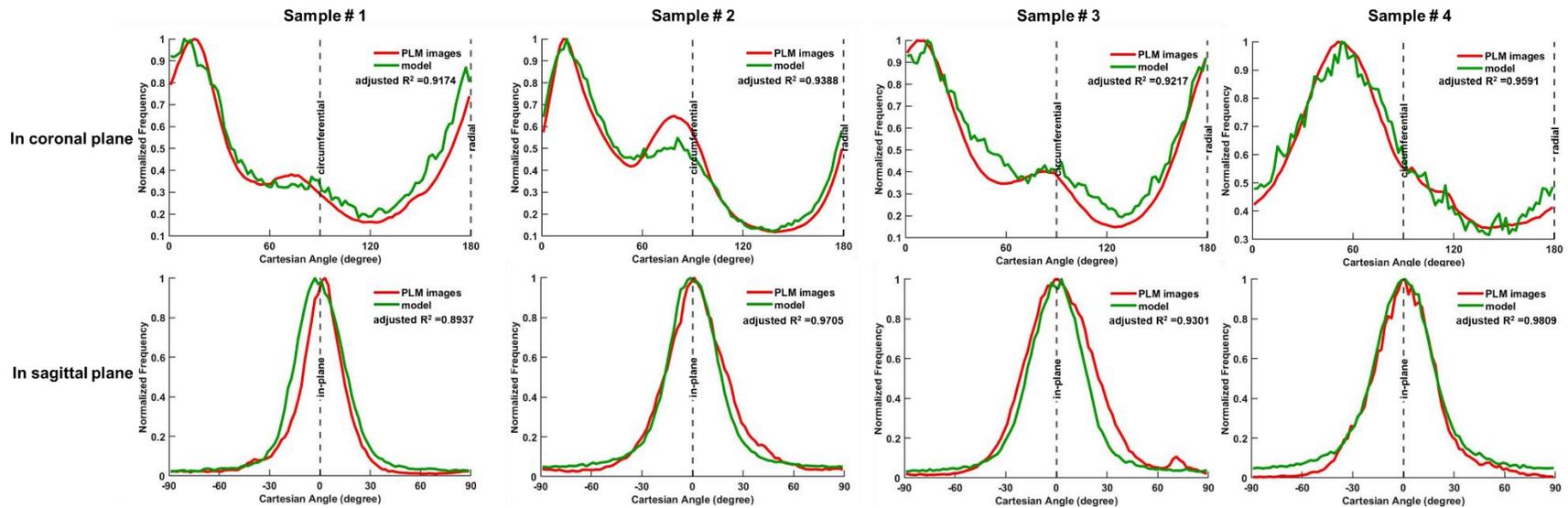
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773 **Figure 3.** Example PLM images and process of image post-processing. **(A)** Example PLM image
 774 of a coronal section. A square-shape block was cropped from the stack and used as a reference
 775 to build the direct fiber model. **(B)** Example PLM image of a sagittal section. The model's in-depth
 776 fiber orientation distribution was adjusted based on the orientations obtained from the yellow block.
 777 The position of this block corresponded to the position of the square-shaped block obtained from
 778 the coronal section. The colors indicate the local fiber orientation in the section plane, with
 779 brightness proportional to the "energy" parameter. **(C)** The original stack showed irregular
 780 thickness from one location to another. **(D)** After interpolation, the thickness of the stack was
 781 uniformized which facilitates the tracing of fibers during the construction of the direct fiber model.



782

783 **Figure 4.** Isometric view of an example **(A)** direct fiber model and **(B)** matrix model. **(C)**
 784 Displacement boundary conditions were applied to the fiber matrix assembly, with components u ,
 785 v representing displacement in X (radial) and Y (circumferential) direction, respectively. The
 786 displacement in Z direction was not constrained, given that fish hooks and strings did not restrict
 787 the sample's displacement in the Z direction. In the first step biaxial stretch, the value of ϵ_{radial} and
 788 $\epsilon_{\text{circumferential}}$ were optimized and derived in the inverse fitting procedure. In the second step biaxial
 789 stretch, the experimental strain values were assigned to ϵ_{radial} and $\epsilon_{\text{circumferential}}$, aiming to match the
 790 model with the stress-strain behaviors observed in the experimental data.

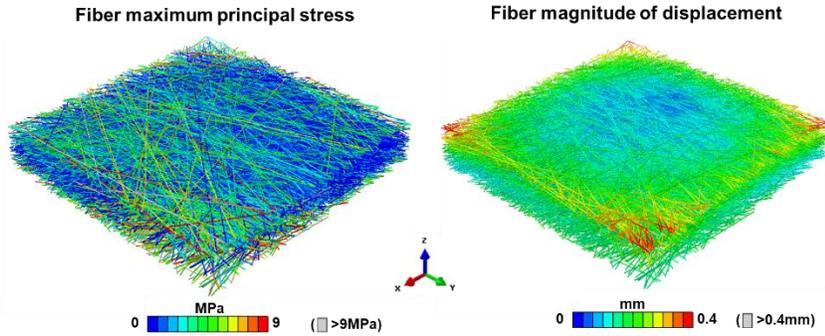


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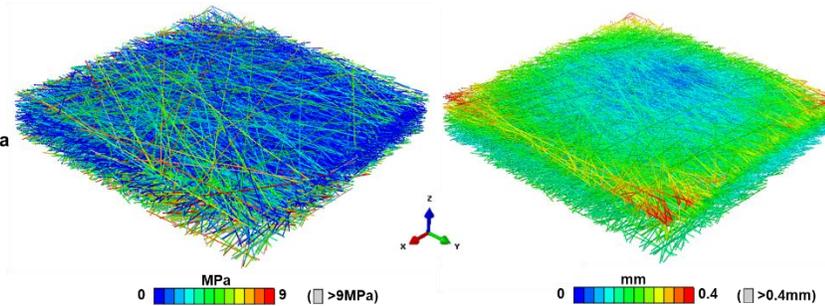
792 **Figure 5.** Fiber orientation distributions of the direct fiber model (green lines) and the corresponding PLM images (red lines). The
 793 analysis was performed in both the **(top row)** coronal and **(bottom row)** sagittal planes of the four samples. In the coronal plane, the
 794 PLM orientation was determined by analyzing a stack of all coronal images, with the radial direction represented by 0 and 180°, and
 795 the circumferential direction represented by 90°. The sagittal plane displayed the average orientation distribution obtained from two
 796 sections, with the in-plane direction represented by 0°. The frequencies of fiber orientations have been normalized by the total sum of
 797 frequencies for effective comparison. The results demonstrate a strong agreement between the fiber orientation distributions of the
 798 direct fiber model and those observed in the PLM images, in both the coronal and sagittal planes. All the adjusted R^2 values, which
 799 exceeded 0.89, indicate a high level of similarity between the model and experimental data.

X direction: radial direction
Y direction: circumferential direction

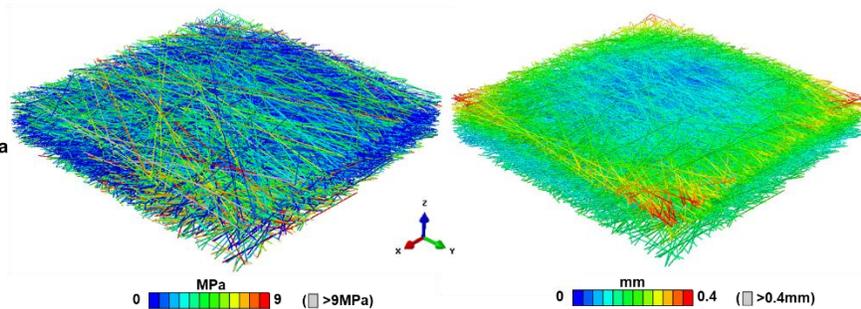
Loading protocol 1:1
radial 150 kPa – circumferential 150 kPa



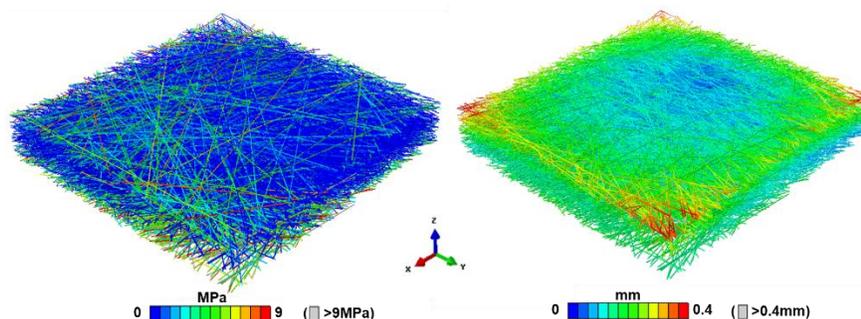
Loading protocol 1:0.75
radial 150 kPa – circumferential 112.5 kPa



Loading protocol 0.75:1
radial 112.5 kPa – circumferential 150 kPa

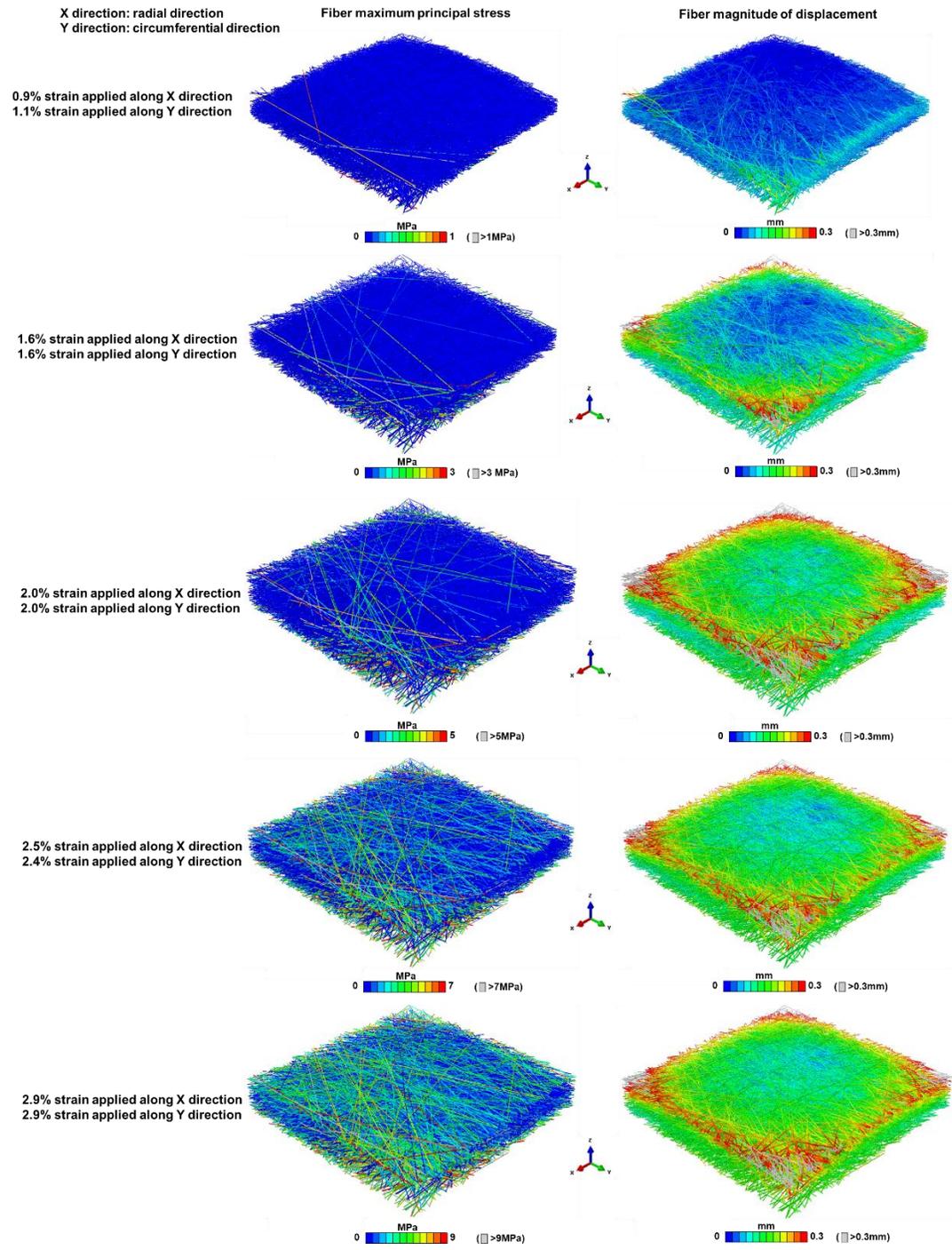


Loading protocol 1:0.5
radial 150 kPa – circumferential 75 kPa



800

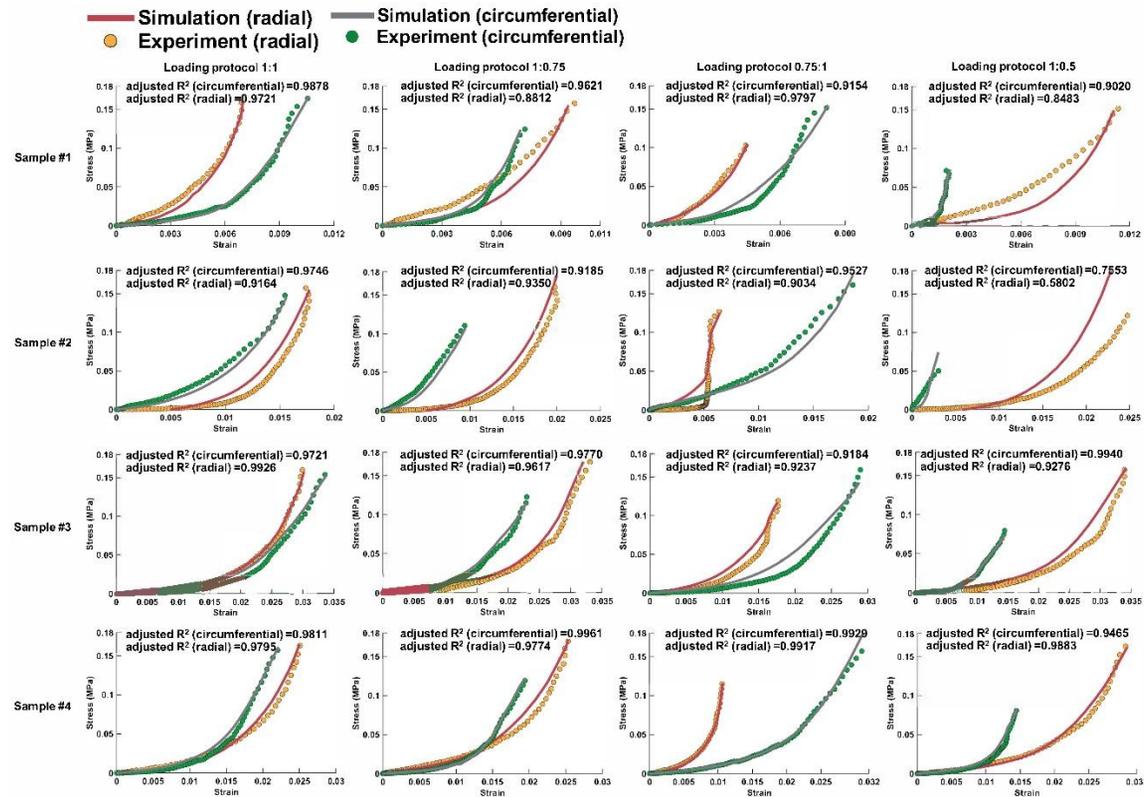
801 **Figure 6.** Isometric views of the direct fiber model of Sample #4 at full stretch. The visualization
802 was enhanced by coloring the fibers based on two important mechanical parameters: the
803 maximum principal stress (shown in the left column) and the magnitudes of displacement (shown
804 in the right column) for each loading protocol. The variations in stress and displacement patterns
805 can be observed, highlighting the non-uniform distribution of stresses and displacements within
806 the tissue.



807

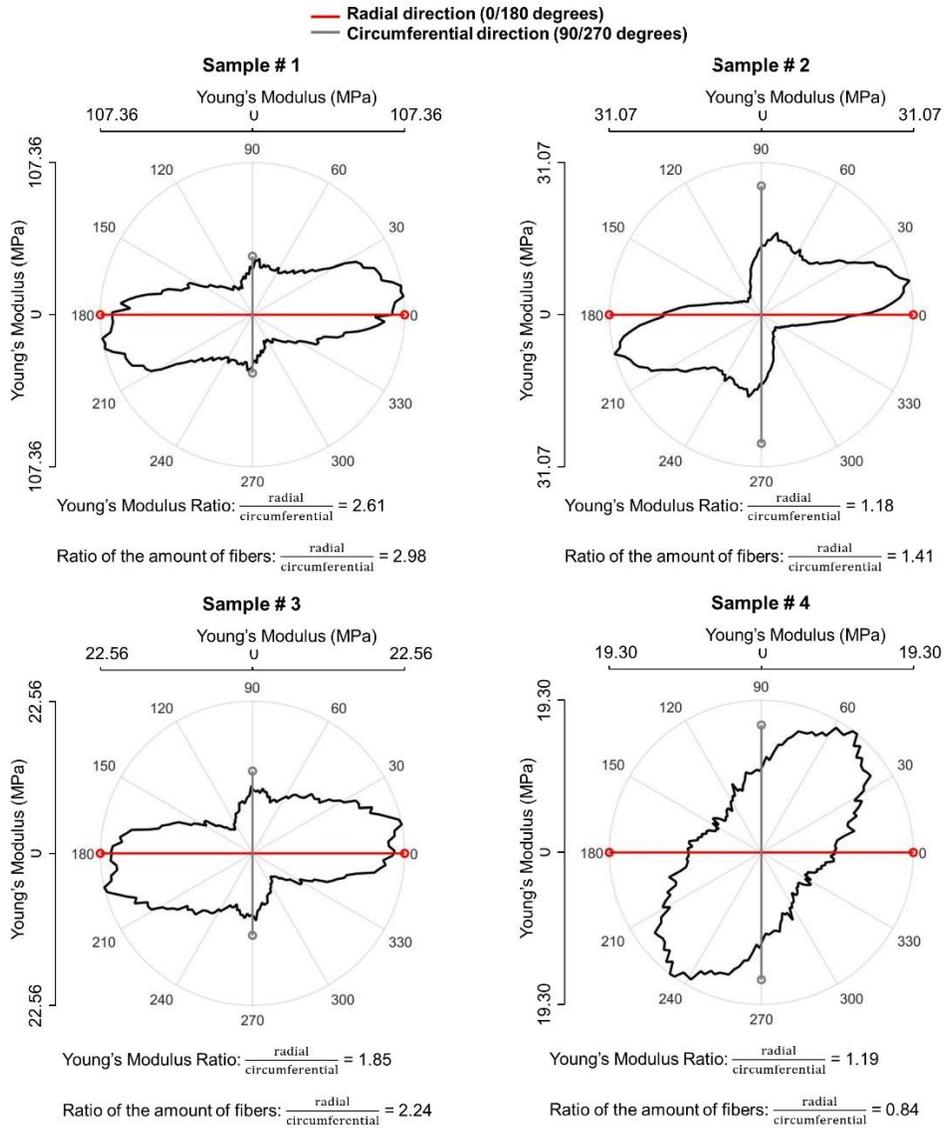
808 **Figure 7.** Isometric views of the direct fiber model of Sample #4 at different stages of the
 809 simulation while undergoing loading protocol 1:1. The visualization was colored based on the
 810 maximum principal stress (left column) and the magnitudes of displacement (right column). In the
 811 early stage of the simulation, the model exhibited some curvature. As it underwent stretching, the

812 model gradually transformed into a flattened configuration. As stretching was applied, a larger
813 number of fibers experience higher levels of stress.



814

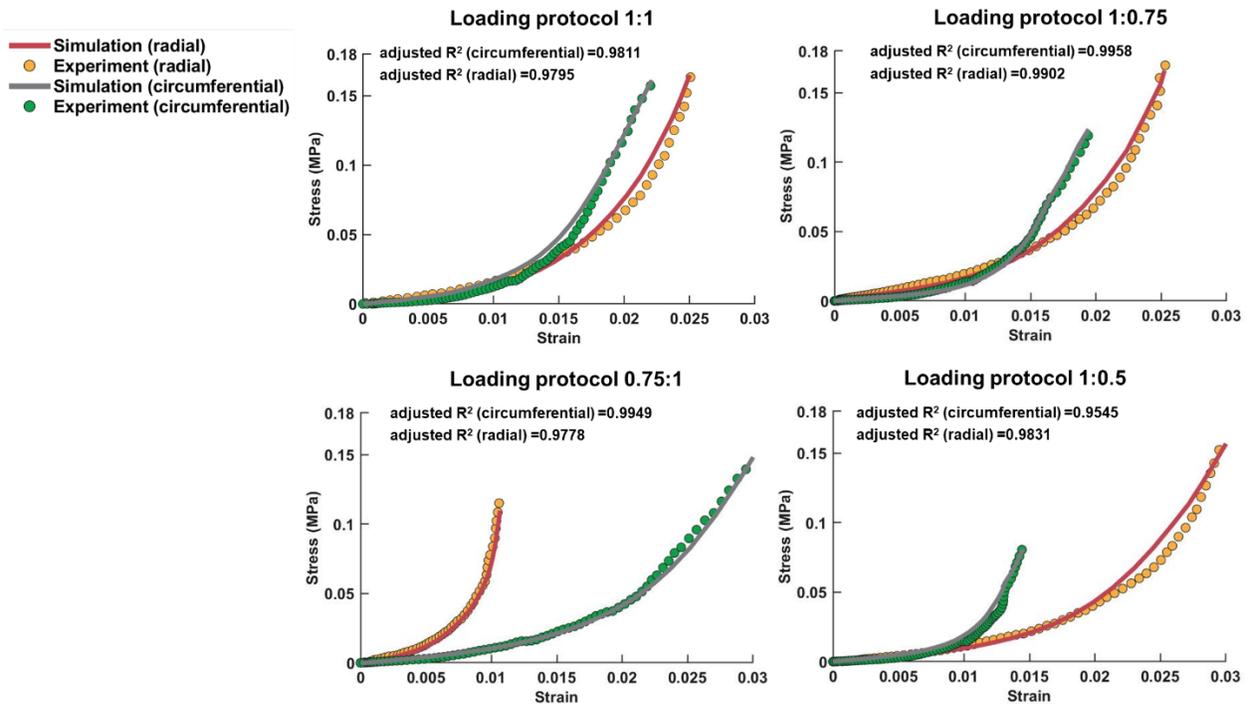
815 **Figure 8.** Stress-strain responses of the fiber-matrix assembly and the corresponding experimental data. The stress-strain curves were
 816 presented for both the radial and circumferential directions in each of the loading protocols. The fiber material properties were
 817 determined by fitting the model to the stress-strain data of loading protocol 1:1 (first column). These derived material parameters were
 818 directly used to predict the response of other loading protocols. The results demonstrated a successful fit between the model and the
 819 experimental data, with consistent agreement achieved simultaneously in both the radial and circumferential directions for each loading
 820 protocol. The goodness of fit was quantified using the adjusted R² value, which exceeded 0.9 in most of the cases. Notably for each
 821 sample, the stress-strain responses were obtained using the same fiber material properties throughout the simulations, with the
 822 variations observed solely in the pre-stretching strains.



823

824 **Figure 9.** Orientation distribution and mechanical anisotropy of all the models. The orientation
 825 distributions are depicted in polar plots. The red and gray solid lines indicate the Young's moduli
 826 in the radial and circumferential directions, respectively, calculated from the slope of the model's
 827 stress–strain response at the maximum strain. All stiffness values are plotted using the same
 828 modulus scale for direct comparison. Below each plot, two anisotropy metrics are reported: 1) the
 829 Young's modulus ratio (radial/circumferential), representing mechanical anisotropy, and 2) the
 830 fiber amount ratio (radial/circumferential), representing structural anisotropy based on fiber count
 831 and orientation frequency (from Figure 5). The results show that while fiber architecture influences
 832 stiffness, the relationship is not strictly proportional, indicating other structural mechanisms may
 833 contribute to anisotropic behavior.

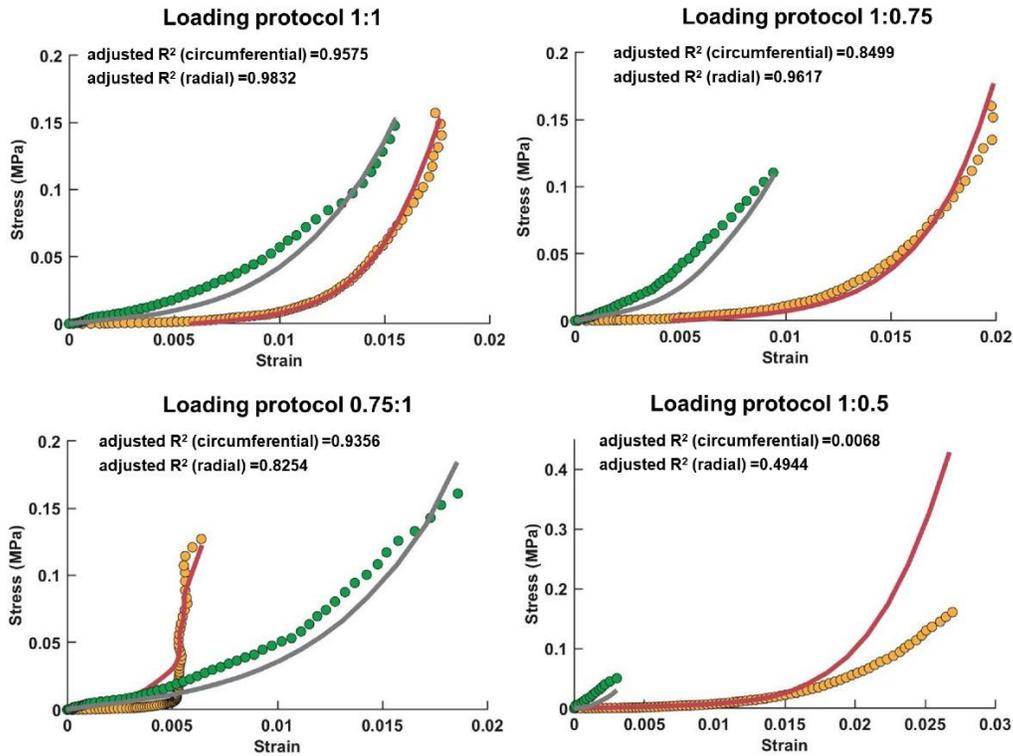
Loading protocol #	C_{10}	C_{01}	Pre-stretching strain (radial)	Pre-stretching strain (circumferential)	Fiber elastic modulus (GPa)	Fiber bulk modulus (GPa, $2(C_{10} + C_{01})$)
1:1	86.20	-0.049	0.0041	0.0074	0.512	0.172
1:0.75	97.601	-0.976	0.0045	0.005	0.574	0.193
0.75:1	77.729	-0.777	0.0135	0	0.457	0.154
1:0.5	86.625	-1.055	0.002	0.005	0.508	0.171



834

835 **Figure 10.** This figure illustrates the findings from the analysis using Sample #4 to assess the
836 similarity of derived material properties when fitting the model to experimental data from different
837 loading protocols. The table demonstrates that the values of C_{10} and C_{01} derived from fitting the
838 model to different loading protocols are closely aligned, resulting in comparable fiber elastic
839 modulus and fiber bulk modulus. Additionally, the stress-strain curves between the model and
840 experimental data exhibit a strong overall fit, with an adjusted R^2 greater than 0.9. These results
841 support the conclusion that the choice of loading protocol for fitting does not influence the derived
842 fiber material properties.

	Loading protocol #	C_{10}	C_{01}	Pre-stretching strain (radial)	Pre-stretching strain (circumferential)
— Simulation (radial)	1:1	581.35	-0.187	0.002	0.007
● Experiment (radial)	1:0.75			0.001	0.01
— Simulation (circumferential)	0.75:1			0.012	0.005
● Experiment (circumferential)	1:0.5			0	0



843

844 **Figure 11.** This figure presents the results of using the material parameters derived from Sample
845 #1 to simulate the experimental stress-strain behavior of Sample #2, using Sample #2's fiber
846 structure, aimed at assessing the possibility of achieving an overly favorable match between the
847 models and experimental data. Following the process described in the main text, the fiber material
848 properties C_{10} and C_{01} were derived by fitting the model to loading protocol 1:1. The results
849 indicate successful matches between the model and experimental data for loading protocols 1:1,
850 1:0.75, and 0.75:1, with overall adjusted R² values exceeding 0.8. However, for loading protocol
851 1:0.5, the model exhibits softer behavior than the experiment in the circumferential direction and
852 stiffer behavior in the radial direction, resulting in adjusted R² values lower than 0.5. As such, it
853 cannot be considered as a valid match. These findings reinforce that the model's applicability is
854 not universal but rather dependent on accurately matching the orientation distribution of the
855 specific sample in order to replicate its experimental data.

Loading Protocol #	Radial (kPa)	Circumferential (kPa)
1:1	150	150
1:0.75	150	112.5
0.75:1	112.5	150
1:0.5	150	75
0.5:1	75	150

856 **Table 1.** Maximum radial (anterior-posterior) and circumferential (equatorial) stress values for
857 each biaxial loading protocol.

Sample #	Temporal/Superior	Sheep #	Thickness(μm)	Number of Coronal Sections
1	Superior	Eye-1	528	50
2	Temporal	Eye-2	1168	87
3	Superior	Eye-2	635	60
4	Temporal	Eye-3	838	74

858 **Table 2.** Information about the four posterior sclera samples used for direct fiber modeling. The
859 samples were obtained from both the superior and temporal quadrants of three sheep eyes. Each
860 sample had a different thickness, resulting in a variation in the number of collected coronal
861 sections.

Sample #	Loading protocol #	C_{10}	C_{01}	Pre-stretching strain (radial)	Pre-stretching strain (circumferential)	Fiber elastic modulus (GPa)
1	1:1	4030	-2.0150	0.00825	0.00475	23.97
	1:0.75			0.00600	0.00725	
	0.75:1			0.00950	0.00750	
	1:0.5			0.00450	0.01050	
2	1:1	542.7	-0.1740	0.00050	0.00775	3.22
	1:0.75			0	0.01	
	0.75:1			0.01000	0.00625	
	1:0.5			0	0	
3	1:1	78.98	-0.058	0	0.00400	0.47
	1:0.75			0	0.01100	
	0.75:1			0.01	0.01000	
	1:0.5			0	0.01350	
4	1:1	86.20	-0.0485	0.00410	0.00740	0.51
	1:0.75			0.00600	0.00510	
	0.75:1			0.01300	0	
	1:0.5			0.00150	0.00550	

862 **Table 3.** The optimized fiber material properties (C_{10} and C_{01}) and the pre-stretching strains along radial and circumferential direction

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